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# Effect of Curing Temperature on Nano-Silver Paste Ink for Organic Thin-Film Transistors

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Silver (Ag) metal electrode having 20  $\mu$ m channel length was printed by reverse offset printing (ROP) using nano-silver paste ink for the source/drain of organic thin-film transistors (OTFT). Specific resistance and surface roughness of printed Ag electrodes with increasing curing temperature were investigated, and surface morphology and grain growth mechanism were systematically verified using a scanning electron microscope (SEM) and atomic force microscope (AFM) in order to obtain an optimized ROP Ag electrode. The Ag electrode was applied to fabricate top-gate/bottom-contact poly(3-hexylthiophene) OTFT devices, which showed reproducible OTFT characteristics such as the field-effect mobility, threshold voltage, and an on/off-current ratio of  $\sim 10^{-3}$  cm<sup>2</sup>/Vs, 0.36 V, and  $\sim 10^2$ , respectively.

Keywords: Curing Temperature, Nano-Silver Paste, OTFT, Reverse Offset Printing.

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# 1. INTRODUCTION

The field of organic electronics has been widely studied by many researchers wanting to develop a variety of high performance and/or high resolution organic opto/electronic devices, such as organic light-emitting diodes (OLEDs),<sup>1</sup> organic field-effect transistors (OFETs),<sup>2,3</sup> organic photovoltaics (OPVs),4 organic non-volatile memory,5 and bio-/chemical-sensors.<sup>6</sup> Various graphic art printing techniques, such as gravure, inkjet, and reverse offset printing (ROP) have been shown to make it possible to realize flexible, large-area, and low-cost organic opto/electronic devices and circuits. For instance, Baeg et al. developed high speed complementary polymer circuits of  $\sim$ 50 kHz switching speeds based on the inkjet-printing process;<sup>7</sup> however, they used conventional photolithographic metal electrode processing for fabrication of the organic thin-film transistors (OTFTs).

The mass-production printing methods, such as gravure and ROP, are substantially notable for largely enhanced productivity and throughput in comparison with other printing methods.<sup>8</sup> However, printing resolution and surface morphologies of the printed patterns via massproduction roll-to-roll (R2R) processes have limited the application area of such printing processes. State-of-theart printing resolution is in the range of 20  $\mu$ m for ROP and gravure printing, with 1  $\mu$ m below film thickness. It is well known that the improvement of charge carrier mobility and the downscaling of channel length (*L*) are essential in order to increase switching speeds of OTFTs and their integrated circuits.<sup>9</sup> Gravure printing, however, has typically been limited to very low device performance because of its poor line resolution and registration.

In this study, therefore, the ROP technique was used for fabrication of OTFTs. It should be noted that this printing method is very attractive because of a variety of advantages, such as very high resolution (potentially available up to below 1  $\mu$ m) and high throughput, no limitation to substrate types or sizes, and ability to print very tiny and various shaped patterns. Surface morphology and electrical properties of the printed nano-silver paste ink after changing the post thermal annealing process were systematically investigated. Optimized Ag printed patterns were utilized to fabricate top-gate/bottom-contact polymer transistors that showed reproducible and acceptable OTFT characteristics.

#### 2. EXPERIMENTAL DETAILS

An Ag electrode having 20  $\mu$ m channel length was fabricated using a sheet-to-sheet reverse offset printer (Narae

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Fig. 1. Schematic illustration of the plate to plate reverse offset printing process.

NanoTech Corporation). The nano-silver paste ink (DGH ink, Advanced Nano Products Co., LTD.) (39 wt%) with 1.5 cPs of viscosity and 25.8 mN/m of surface tension was dispensed on the surface of the blanket roll (PDMS type, KNW) by controlling the syringe pump. While the blanket rolled on cliché with 20 µm patterns at 15 mm/s speed, unnecessary ink was removed from the blanket and was transferred to the top of the cliché surface. The remained ink for the desired pattern on the blanket was transferred onto the  $160 \times 160$  mm glass substrates at room temperature (Fig. 1). Post thermal treatment was individually carried out in the furnace at temperatures ranging from 150 to 450 °C with 50 °C steps for 30 minutes in order to remove the various additives and residual solvent in the printed Ag ink. The sheet resistance and thickness of the printed Ag patterns were measured at about five different points. The crystal structure and the surface morphology of the printed Ag ink were analyzed using a scanning electron microscope (SEM) and atomic force microscope (AFM) in order to optimize curing temperature of the metal electrodes. The poly(3-hexylthiophene) (P3HT) semiconducting polymer was dissolved in *p*-xylene (10 mg/ml), and poly(methyl methacrylate) (PMMA) gate dielectric material was dissolved in n-butyl acetate (80 mg/ml) at 80 °C for more than two hours. These solutions were subsequently spincoated upon the printed Ag electrode. Finally, the Al gate electrode (100 nm) was deposited by thermal evaporator. I-V characteristics of the OTFT were measured using a KEITHLEY 4200-SCS in the dark in air.

## 3. RESULTS AND DISCUSSION

Figure 2 shows an optical microscope image of the printed source/drain (S/D) electrodes fabricated by ROP; these electrodes had a channel length of 20  $\mu$ m. The thickness of the printed Ag electrodes was about  $400 \pm 20$  nm. High resolution electrode patterns, with  $L = 20 \ \mu$ m and  $W = 80 \ \mu$ m, were typically obtained; individual OTFT were treated post annealing at 150, 200, 250, 300, 350, 400 and 450 °C, each for 30 min.

The specific resistance, which was calculated from the thickness and the sheet resistance of the printed Ag pattern, was found to decrease rapidly with the increasing of the curing temperatures until 350 °C; thereafter it increased slightly above the critical point (Fig. 3(a)). The value of



Fig. 2. Optical microscope image of the source and drain electrode made by reverse offset printing.

the specific resistance of the printed Ag ink was ca.  $1.0 \times$  $10^{-5}$  ohm  $\cdot$  cm at 150 °C, but rapidly dropped to 2.8  $\times$  $10^{-6}$  ohm  $\cdot$  cm at 350 °C and increased to  $4.5 \times 10^{-6}$  ohm  $\cdot$ cm at temperatures above 350 °C. The decreased specific resistance with increasing curing temperature was mainly attributed to the growth of Ag grains and to the complete removal of the various additives and residual solvent in the printed Ag patterns. However, the specific resistance was not further reduced with the increased curing temperature above the critical annealing point (350 °C). This can be explained by considering the SEM surface images of the printed Ag ink for curing temperature, as shown in Figure 3(b). It can be clearly observed that the grain size increased gradually with the increased curing temperature. This expanation is in good agreement with the experimental results shown in Figure 3(a). The initial crystal components of the nano-silver paste ink were found to impinge on each other in the two-dimensional orientation and their placement led them to adhere and form the crystal. Finally, the crystal structure grew as a whole, including in the three-dimensional orientation. Beyond the driving force that level of supersaturation, and discontinous and cracked surface were found to occur, as shown in the inserted SEM images, at more than 400 °C. These defects, such as the discontinous and rough surface, are likely to bring a decrease in conductance of the printed Ag ink. In this regard, we can find that surface morphology was rapidly getting rough over 350 °C, as shown in Figures 4(a and b). Figure 4 shows the surface roughness and the AFM morphology of the printed Ag ink. The root mean square roughness  $(R_{\rm rms})$  tends to increase, accompanying the grain growth from 150 to 450 °C, but the  $R_{\rm rms}$  was significantly increased from 29.1 nm at 350 °C to 72.0 nm at 450 °C.

The OTFT with reverse-offset-printed S/D electrodes was fabricated, of which the  $L = 20 \ \mu \text{m}$ ,  $W = 80 \ \mu \text{m}$  and thickness =  $400 \pm 20 \ \text{nm}$  (Fig. 5). In our experiment,

450



**Fig. 3.** (a) Specific resistance and (b) corresponding scanning electron microscope images of the printed nano-silver paste after each thermal annealing process.

in order to fully cover the relatively thick and rough source/drain electrodes, a thick active layer and insulator, as much as electrodes, was necessary for the top-gate/bottom-contact transistors. Because a smooth surface of the printed electrode is desirable for the high performance of the OTFT device, continuous aticve layers were deposited on top of the Ag patterns. For the active layer, the P3HT, which has ionization potential of -4.9 eV, was used as a polymer semiconductor.<sup>10</sup> As insulator film, PMMA, with more than 500 nm thickness ( $C_i = 6.2 \text{ nF/cm}^2$ ), was spin-coated onto the active layer. The Al was deposited by thermal evaporator on the top layer as a top gate metal. The transfer and output characteristics of these

Fig. 4. (a) Root mean square roughness and (b) corresponding surface morphologies, measured by atomic force microscope of the printed source and drain electrode.

OTFT are shown in Figures 6(a and b). The calculated field-effect mobility of the OTFT in the saturation region was ca.  $3 \times 10^{-3}$  cm<sup>2</sup>/Vs. It should be noted that the maximum mobility of the P3HT OTFT device was more than  $1 \times 10^{-1}$  cm<sup>2</sup>/Vs when measured in the inert atmosphere on a photolithography patterned gold source/drain electrode.<sup>7</sup> Although the mobility is relatively lower than that of the top-gate reference devices, it is still comparable



Fig. 5. Atomic force microscope image of the printed Ag source and drain electrodes made by reverse offset printing.



**Fig. 6.** (a) Transfer and (b) output characteristics of OTFT with printed source and drain electrodes.

to most of the typical bottom-gate P3HT OTFTs, as well as those made by using the Ag electrode. The threshold voltage and the on/off-current ratios were 0.36 V and under  $10^2$ , respectively. This relatively low on/off ratio can be attributed to the spin-coated P3HT active layer (unpatterned) and to the measurement in air, because the semiconducting polymer P3HT has low oxidation stability.<sup>11</sup> Because of the work-function mismatch between Ag (-4.2 eV) and the highest occupied molecular orbital (HOMO) of *p*-type polymer semiconductor (-4.9 eV),<sup>10</sup> we believe that the relatively low field-effect mobility of the printed Ag source/drain transistors presumably resulted from the contact resistance. Also, the rough surface and some of the large peaks of printed electrode led to relatively high gate leakage current because of defects between the active layer and the gate insulator. This is a common problem for many printed electrode materials. Research into reducing contact resistance and gate leakage current, and thereby optimizing the OTFT performance, is currently under-way. We expect that high resolution and high throughput roll-to-roll printable Ag metal electrodes can be utilized in a variety of printed electronic applications, such as printed RFID tags,<sup>12</sup> TFT-backplanes for active matrix LCD and OLED<sup>1–3</sup> displays, and platforms of bio- and optical-sensors.<sup>6</sup>

# 4. CONCLUSION

We have demonstrated printed Ag source/drain with 20  $\mu$ m short channel length by reverse offset printing, and investigated the effect of curing temperature on nano-silver paste ink for transistor electrodes. Reproducible and acceptable characteristics of top-gate OTFTs were achieved. But the printing process yields surface roughness and too thick of an electrode, etc., and many other problems remain.

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